# Synthesis and Crystal Structure of a New One-dimensional Quaternary Thiophosphate, CsNb<sub>2</sub>PS<sub>10</sub>

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Group 5 transition metal thiophosphates form an interesting class of materials due to their low-dimensional structural varieties and interesting anisotropic properties.<sup>1</sup> Especially, these low-dimensional compounds have been extensively investigated for potential applications as cathode materials of high energy density secondary batteries.<sup>2</sup> Preparation of the alkali metal thiophosphates is crucial to understand the nature of the intercalation and to analyze their behavior during charge and discharge processes. This gives us motivation to insert alkali metals into the thiophosphates. However, no stable alkali metal intercalated product in this system has been obtained and structurally characterized. We believe that the crystallinity of the host structure is not maintained during the electrochemical insertion procedure. During the past decade we have applied halide flux method to bypass this problem. This technique exploits metal halides as a flux and as a reactant. The resultant products usually include various monovalent cations and this technique enabled us to find the first niobium thiophosphates containing alkali metals, KNb<sub>2</sub>PS<sub>10</sub>.<sup>3</sup> During an effort to expand the range of compounds within the quaternary alkali metalgroup 5 metal-P-S system by substituting alkali metals, we were able to prepare single crystals of CsNb<sub>2</sub>PS<sub>10</sub>. Here we report the synthesis and characterization of a new quaternary thiophosphate, CsNb<sub>2</sub>PS<sub>10</sub>.

The structure of CsNb<sub>2</sub>PS<sub>10</sub> is closely related to those of the previously reported group 5 metal thiophosphates,  $KNb_2PS_{10}$ <sup>3</sup>  $RbNb_2PS_{10}$ <sup>4</sup>  $AgNb_2PS_{10}$ <sup>5</sup>  $AuNb_4P2S_{20}$ <sup>6</sup> and  $NaNb_2PS_{10}$ <sup>5</sup> It consists of the one-dimensional  $\[mu]{}^1[Nb_2PS_{10}^-]$ chains along the [001] direction (Fig. 1), and these anionic chains are stabilized by electropositive Cs<sup>+</sup> ions. A stereoscopic view of the unit cell of CsNb<sub>2</sub>PS<sub>10</sub> is shown in Figure 2. Like most of the group 5 metal thiophosphates, the chains found in CsNb<sub>2</sub>PS<sub>10</sub> are composed of the typical biprismatic [Nb<sub>2</sub>S<sub>12</sub>] and [PS<sub>4</sub>] units. Two crystallographically distinct, but chemically identical chains are found in this structure. The Nb atoms are surrounded by 8 S atoms in a bicapped trigonal prismatic fashion. Two prisms are sharing a rectangular face to form the  $[Nb_2S_{12}]$  unit (Fig. 3). This unit is similar to those found in  $ANb_2PS_{10}$  (A=K, Rb)<sup>3,4</sup> in the arrangement of the (S-S)<sup>2-</sup> ligands. Although the ligands occupy the same sites, a 2-fold rotation symmetry is found in ANb<sub>2</sub>PS<sub>10</sub>,<sup>3,4</sup> whereas an inversion symmetry is found in



**Figure 1.** A view of  $C_8Nb_2PS_{10}$  showing an individual  ${}_{\infty}^{-1}[Nb_2PS_{10}^{-1}]$  chains and coordination around Nb and P atoms. Displacement ellipsoids are drawn at the 60% probability level. The labeling scheme is shown. Large and small filled circles are Cs and Nb atoms respectively; open circles and grey circles represent S and P atoms respectively. The Cs-S bonds are omitted for clarity.

 $A_xAg_{0.5}Nb_2PS_{10}$  (A=K, Rb).<sup>7,8</sup> These units are bound through the S-S prism edges and through one of the capping sulfur atoms to make infinite  ${}_{\infty}^{1}[Nb_2S_9]$  chains. One of the S atoms at the prism edge and two other capping S atoms are bound to the P atom and an additional S atom is attached to the P atom to complete the infinite  ${}_{\infty}^{1}[Nb_2PS_{10}^{-}]$  chains. The P-S distances are in good agreement with the P-S bonding distances found in other thiophosphates.<sup>9-13</sup> Atoms S7 and S16 are terminal atoms and this is responsible for the short distances (P1-S7 1.973(6) Å; P2-S16 1.971(6) Å) and their large displacement parameters.

In the arrangement of Nb atoms, Nb ... Nb interactions



Figure 2. A stereoscopic view of the unit cell of CsNb<sub>2</sub>PS<sub>10</sub>.



Figure 3. A perspective view of the  $[Nb_2S_{12}]$  units with approximate 2-fold symmetry. Small filled circles are Nb atoms and large open circles are S atoms. Nb-S bonds are omitted except for the capping S atoms for clarity.

alternate in the sequence of one short (Nb1-Nb2, 2.884(2) Å or Nb3-Nb4, 2.891(2) Å) and one long (Nb1-Nb2, 3.764(2) Å or Nb3-Nb4, 3.758(2) Å). The short distances are typical of Nb<sup>4+</sup>-Nb<sup>4+</sup> bond<sup>14</sup> and the long Nb<sup>++</sup>Nb distances show that there is no significant intermetallic bonding interaction and such an arrangement is consistent with the high resistivity of the compound.

There are no interchain interactions except the van der Waals forces. Instead, the Cs<sup>+</sup> cations reside in the van der Waals gap. They are surrounded by eight S atoms if we arbitrarily choose a cutoff of 3.9 Å for the Cs<sup>···</sup>S. Thus, the classical charge balance of the compound can be represented by  $[Cs^+][Nb^{4+}]_2[PS_4^{3-}][S_2^{2-}]_3$ . Studies to solubilize NaNb<sub>2</sub>PS<sub>10</sub> with polar organic solvents such as N-methylformamide (NMF) to obtain nanotubules based on polymeric  ${}_{\infty}^{1}[Nb_2PS_{10}^{-}]$ chains have been reported<sup>14</sup> and we believe that CsNb<sub>2</sub>PS<sub>10</sub> would be ideal for this purpose because of the highly electropositive nature of the Cs<sup>+</sup> cation.

## **Experimental Section**

**Synthesis.** The compounds CsNb<sub>2</sub>PS<sub>10</sub> was prepared by the reaction of elemental Nb, P, and S with the use of the reactive halide flux technique. Stoichiometric combinations of the pure elements, Nb powder (CERAC 99.8%), P powder (CERAC 99.5%), and S powder (Aldrich 99.999%)

were mixed in a fused silica tube with the addition of eutectic mixture of CsCl/LiCl. The mass ratio of reactants and fluxes was 1:2. The tube was evacuated  $(10^{-2} \text{ Torr})$ , sealed, and heated gradually to 973 K, where they were kept for 7 days. The tube was cooled to room temperature at the rate of 4 K/hr. The excess halide fluxes were removed with distilled water and dark red needle-shaped crystals were obtained. The crystals are stable in air and water. Analysis of this compound with the microprobe of an EDAX-equipped AMRAY 1200C scanning electron microscope indicated the presence of Cs, Nb, P, and S. No other element was detected.

**X-ray Crystallography.** The crystal structures of CsNb<sub>2</sub>PS<sub>10</sub> was determined by single-crystal X-ray diffraction methods. Preliminary examination and data collection were performed on an MXC3 diffractometer (Mac science) equipped with graphite-monochromatized MoK $\alpha$  radiation ( $\lambda = 0.7107$  Å). Cell constants and an orientation matrix for data collection were obtained from the least-squares analysis, using the setting angles of 22 reflections in the range 20.0° <  $2\theta$ (MoK $\alpha$ ) < 28° that had been automatically centered. The intensities of two standard reflections measured every 100 reflections showed no significant deviations during the data collection. Intensity data for the title compound were collected at 200(2) K with the  $\omega$ -2 $\theta$  scan technique. Additional crystallographic details are described in Table 1. The observed Laue symmetry and the systematic extinction

Notes

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<b>Table 1</b> . Crystal Data and Structure Refinement		Table 3. Selected Bond Lengths [A] and Angles [deg]			
Empirical formula	CsNb <sub>2</sub> PS <sub>10</sub>	Nb1-S13	2.466(5)	Nb3-S3	2.462(5)
Formula weight	670.30	Nb1-S12	2.474(4)	Nb3-S2	2.470(4)
Temperature	200(2) K	Nb1-S14	2.507(5)	Nb3-S4	2.516(5)
Wavelength	0.71073 Å (MoK <i>a</i> )	Nb1-S11	2.516(5)	Nb3-S1	2.534(4)
Crystal system	Monoclinic	Nb1-S18	2.541(4)	Nb3-S8	2.547(4)
Space group	Pc	Nb1-S19	2.545(4)	Nb3-S9	2.554(4)
Unit cell dimensions	a = 14.040(3) A	Nb1-S15	2.562(5)	Nb3-S5	2.566(5)
	$b = 7.544(2) \text{ A } b = 95.72(2)^{\circ}.$	Nb1-S17	2.652(4)	Nb3-S6	2.635(4)
Volume	C = 12.972(3) A 1367 0(5) Å <sup>3</sup>	Nb2-S14 $^i$	2.464(5)	Nb4-S1 <sup><math>i</math></sup>	2.469(4)
Z	4	Nb2-S11 <sup>i</sup>	2.466(4)	Nb4-S4 $^i$	2.488(5)
Density (calculated)	$3.257 \text{ Mg/m}^3$	Nb2-S13 <sup>i</sup>	2.522(5)	Nb4-S $3^i$	2.526(4)
Absorption coefficient	$5.902 \text{ mm}^{-1}$	Nb2-S19	2.550(4)	Nb4-S $2^i$	2.526(5)
Crystal size	$0.70 \times 0.04 \times 0.04 \text{ mm}^3$	Nb2-S12 <sup>i</sup>	2.555(5)	Nb4-S9	2.548(4)
Theta range for data collection	2.70 to 24.99°	Nb2-S18	2.577(4)	Nb4-S8	2.571(4)
Index ranges	$-16 \le h \le 16, -8 \le k \le 0,$	Nb2-S20	2.582(5)	Nb4-S10	2.590(5)
Reflections collected	$0 \le 1 \le 15$ 3339	Nb2-S17	2.646(4)	Nb4-S6	2.647(4)
		Nb1-Nb2 <sup>ii</sup>	2.884(2)	Nb3-Nb4 <sup>ii</sup>	2.891(2)
Independent reflections $C_{\text{opendent}} = 24.00^{\circ}$	2519 [R(int) = 0.0427]	S11-S12	2.038(6)	S1-S2	2.048(6)
Absorption correction $= 24.99^{\circ}$	100.0 %	S13-S14	2.022(7)	S3-S4	2.043(7)
	Meulenar <sup>17</sup> )	S18-S19	2.050(6)	S8-S9	2.064(6)
Data / restraints / parameters	2519 / 2 / 253	P2-S16	1.971(6)	P1-S7	1.973(6)
Goodness-of-fit on F2	1.125	P2-S20	2.046(6)	P1-S5	2.043(6)
Final R indices [I>2sigma(I)]	R1 = 0.0364, wR2 = 0.0808	P2-S15	2.050(6)	P1-S10	2.060(6)
R indices (all data)	R1 = 0.0503, wR2 = 0.0956	P2-S17	2.087(6)	P1-S6	2.087(6)
Absolute structure parameter	-0.02(3)				
Largest diff. peak and hole	1.044 and $-1.236 \text{ e.}\text{\AA}^{-3}$	S16-P2-S20	113.0(3)	S7-P1-S5	112.9(3)
		S16-P2-S15	112.6(3)	S7-P1-S10	112.6(3)
Table 2. Atomic coordinates (	S20-P2-S15	112.1(3)	S5-P1-S10	112.7(3)	

S16-P2-S17

S20-P2-S17

S15-P2-S17

**Table 2.** Atomic coordinates  $(\times 10^4)$  and equivalent isotropic displacement parameters (Å<sup>2</sup>×10<sup>3</sup>)

	х	У	Z	$\mathrm{U(eq)}^{*}$
Cs1	5987(1)	4998(2)	9626(1)	34(1)
Cs2	981(1)	4811(2)	7183(1)	35(1)
Nb1	8712(1)	9436(2)	8597(1)	17(1)
Nb2	8687(1)	9310(2)	5694(1)	16(1)
Nb3	3746(1)	9346(2)	6265(1)	14(1)
Nb4	3719(1)	9389(2)	3366(1)	15(1)
P1	2991(3)	5964(5)	4750(3)	17(1)
P2	7972(3)	5972(5)	7139(3)	20(1)
S1	2503(3)	8611(6)	7489(3)	19(1)
S2	2439(3)	11121(6)	6885(4)	22(1)
<b>S</b> 3	4992(3)	8692(6)	7687(3)	21(1)
S4	4997(3)	11281(7)	7226(4)	25(1)
S5	3901(3)	5974(6)	6082(4)	23(1)
S6	2477(3)	8562(5)	4696(3)	16(1)
<b>S</b> 7	2009(3)	4092(6)	4734(4)	32(1)
<b>S</b> 8	4975(3)	9474(6)	4949(3)	20(1)
S9	4035(3)	11586(5)	4853(3)	18(1)
S10	3701(3)	5959(6)	3435(4)	20(1)
S11	7463(3)	8736(6)	9799(4)	21(1)
S12	7398(3)	11232(6)	9194(3)	20(1)
S13	9999(3)	8899(6)	9999(4)	24(1)
S14	9917(3)	11462(6)	9542(4)	24(1)
S15	8878(3)	6061(6)	8479(4)	22(1)
S16	7003(4)	4085(6)	7157(4)	33(1)
S17	7440(3)	8556(5)	7031(3)	18(1)
S18	9939(3)	9490(6)	7285(3)	18(1)
S19	9001(3)	11578(5)	7146(3)	22(1)
S20	8698(3)	5897(6)	5849(3)	22(1)

\*U(eq) is defined as one third of the trace of the orthogonalized U<sup>ij</sup> tensor.

Symmetry transformations used to generate equivalent atoms: (i) x, -y+2, z-1/2, (*ii*) x, -y+2, z+1/2

S7-P1-S6

S5-P1-S6

S10-P1-S6

115.6(3)

102.2(2)

99.7(2)

115.7(3)

100.1(2)

102.3(2)

(*hOl*: k + l = 2n + 1) were indicative of the monoclinic space group P2/c or Pc. A solution with a low figure of merit could only be obtained in the non-centrosymmetric space group, Pc. The initial positions for all atoms were obtained by using direct methods of the SHELXS-86 program.<sup>15</sup> The structure was refined by full-matrix least-squares techniques with the use of the SHELXL-97 program.<sup>16</sup>

With the composition established the data for the compound were corrected for absorption with the use of the analytical method of Tompa and de Meulenaer.<sup>17</sup> All atomic displacement parameters were refined anisotropically. The final cycle of refinement performed on  $F_0^2$  with 2519 unique reflections afforded residuals wR2 = 0.1023 and conventional R1 index based on the reflections having  $F_0^2 > 2\sigma(F_0^2)$ is 0.0392. However the Flack parameter x = 0.36(4) implied that the absolute structural determination is wrong.<sup>18</sup> Refinement of the inverse structure leads to x = -0.02(3) and the improved reliability factor (wR2 = 0.0956, R1 = 0.0364). A difference Fourier synthesis calculated with phase based on the final parameters shows no peak heights greater than 1.044 e/Å<sup>3</sup> or lower than -1.236 e/Å<sup>3</sup>. No additional sym-

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metry, as tested by PLATON<sup>19</sup> was detected in this structure.

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