## Formation of Self-assembled Nanostructure on Noble Metal Islands Based on Anodized Aluminum Oxide

Jong Bae Park, Youngsic Kim, Seong Kyu Kim,<sup>\*</sup> and Haeseong Lee<sup>†</sup>

Department of Chemistry, Sungkyunkwan University, Suwon 440-746, Korea <sup>†</sup>Jeonnju Branch of Korea Basic Science Institute, Jeonju 561-756, Korea Received October 8, 2003

Key Words : AAO, Aluminum oxide, Electro-deposition, Au, Gold

Anodized aluminum oxide (AAO) has been considered an attractive template for simple fabrication of highly-ordered nanostructures.<sup>1-6</sup> It provides a 2-dimensional array of hexagonal cells with pores of uniform diameter and interpore distance that are adjustable in the range of a few tens to hundreds of nanometers. It can be easily grown on an aluminum sheet with high purity by a sequence of several electrochemical steps; electro-polishing, the 1st anodization, etching, and the 2nd anodization.<sup>7,8</sup> The pores are grown vertically with respect to the AAO surface. The regularity of the pore structure is usually limited by the inherent grain domain in the aluminum sheet to a few micrometers, but can be improved to cover many millimeters of monodomain by pre-indenting the aluminum sheet with SiC<sup>7</sup> or Si<sub>3</sub>N<sub>4</sub> molds.<sup>9</sup> Although fabrication of such molds requires elaborate and costly processes with e-beam nanolithography, such potentially superb regularity can be practically applied to fabrication of nanoscale devices in electronics, optics, biosensors, etc.

If the pores are completely filled with a metal that can bind strongly a functional group of molecules, we can anticipate that further nanoscale network structures can be built on the metal islands exposed over the pores. Filling with a noble metal such as Au or Ag, and binding with self-assembly class of molecules, whose terminal group is either -COOH, -SH, or -NC, would most likely perform this task. This process is schematically summarized in Figure 1.

For this task, we first synthesized the AAO template, whose pore sizes and depths are about 30 nm and 1  $\mu$ m, respectively, following well-known procedures in the literatures.<sup>7,8</sup> Figure 2a show a scanning electron microscope (SEM) image of a typical AAO surface produced in our laboratory. Without removing the base aluminum substrate, we tried to fill the pores with a noble metal, Au in this report. Preserving the aluminum oxide barrier layer between the aluminum substrate and the AAO is necessary since each filled metal has uniform altitude on the barrier.

The electro-deposition of Au (Figure 1b) needs more elaborate additional steps. First, the AAO barrier layer has to be made thinner by a successive drop of the DC voltage<sup>10-12</sup> at the last stage of the 2<sup>nd</sup> anodization process. Then, the AAO/Al is immersed in an Au electroplating solution,

maintained at 60 °C. A commercially available Au electroplating solution (Orotemp24 from Technic Co.) is used. A 10 V AC voltage was applied between the AAO/Al electrode and the counter graphite electrode. Au as a form of rod grows from the bottom of the pores and fills the AAO pores up to the top in about 15 minutes. Figure 2b shows one example of the SEM pictures of the Au-filled AAO surface, which corresponds to the top view of Figure 1b. The bright circular spots shown in the SEM image correspond to the Au columns formed in the pores. The brightness for Au indicates that the SEM electrons impinged onto the Au were not discharged effectively because of the insulating alumina barrier layer. The nonuniform brightness indicates that each Au spot has different height. The cross-sectional view of the grown Au rods (not shown here) indicates that the height fluctuates by 20-30%. The uniformity of the height of the filled metal could be improved if a programmable shaping of the applied voltages upon the electro-deposition were used.<sup>11</sup> However, even with such elaborate efforts, the reported best uniformity in the growth rate is only 10%, far below the level with which any nanostructure can be built on the top surface.11,12

The average diameter of the Au rods in Figure 2b is about 80 nm, over twice the initial average pore diameter. This implies that the inner walls of AAO were etched by the electro-plating solution during the electro-deposition. The unwanted etching of AAO inner walls as well as the nonuniform growth rate seemed beyond the controllable inconvenience and drove us to try an alternative and effective preparation of metal islands over the AAO surface. In the new approach, we take the advantage that each filled metal has the uniform altitude at the bottom side of the AAO pores. The unwanted etching by the electroplating solution is less severe in deeper part of the pores, so the original pore structure is well conserved at the bottom side.

In the next step, the upper surface of the Au-filled AAO was coated with sputtered Pt in order to ensure an electrical contact (Figure 1c). The aluminum substrate is removed with a saturated HgCl<sub>2</sub> solution (Figure 1d). Then, AAO barrier layer was slightly etched by 0.1 M NaOH solution for 10 minutes. In this way the deposited Au rod is exposed slightly (Figure 1e). When the exposed area is small, we were not successful in taking clear SEM images of the resulting surface, since the exposed Au columns had the same altitudes as the surrounding alumina and they were in

<sup>\*</sup>To whom the correspondence should be addressed. e-mail: skkim @skku.edu



Figure 1. Schematics for fabricating AAO-based nanostructures of this work.

contact with the coated Pt (discharging effect). When the AAO barrier layer was over-etched, the exposed Au columns can be seen in the SEM image (Figure 2c). In this picture, the brightness of the exposed Au is almost same as that of the surrounding alumina, but the gap between the two aids the distinction of the two.

Finally, the Au-exposed AAO is dipped into 1 mM solution of 1,4-phenylene diisocyanide (CN-(C<sub>6</sub>H<sub>4</sub>)-NC, Aldrich Co.) for an hour, expecting that the molecule binds exclusively to the exposed area of Au (Figure 1f). This molecule has the two identical terminal groups of -N=C, one



(d)

**Figure 2.** SEM images of AAO-based nanostructures; (a) a typical AAO template, or the top view of Figure 1a. (b) Au-filled AAO, or the top view of Figure 1b. (c) exposed Au column by etching, or the top view of Figure 1e. This is the case when the etching time is longer than desired 10 min. (d) Au nanoparticles bound to the metal islands *via* 1,4-phenylene diisocyanide, or the top view of Figure 1g. The scale bars of (a), (b), (d) are 200 nm. The scale bar of (c) is 100 nm.

of which is chemisorbed on the exposed Au islands. It easily forms a self-assembled monolayer structure on gold surface with the other terminal -N≡C ready to bind other noble metals.<sup>13</sup> We demonstrate this by dipping the self-assembled

Notes

## Notes

array of 1,4-phenylene diisocyanide into the solution containing Au nanoparticles synthesized separately (Figure 1g). The Au nanoparticles were synthesized by reacting 50 mL of 2 mM KAuCl<sub>4</sub> and 5 mL of 38.8 mM sodium citrate dihydrate at 100 °C. The average diameter of the synthesized gold nanoparticles was identified by TEM to about 30 nm. Figure 2d shows the SEM image of the gold nanoparticles sitting exclusively on the central part of the AAO unit cell. The bright spots are absent when the gold nanoparticles are not supplied by dipping, so they must correspond to the gold nanoparticles. The images of the exposed Au islands are not brighter than the surrounding alumina since the filled Au is in contact with the sputtered Pt. In contrast, the gold nanoparticles are displayed in bright images since the selfassembled 1,4-phenylene diisocyanide layer works as an insulating layer.

We have observed this phenomenon many times and recorded over 20 SEM pictures similar to Figure 2d. Based on the analysis of those SEM pictures, we have confirmed the following general trends: (1) the gold nanoparticles tend to cluster together, (2) but they never form overlayers, or one gold nanoparticle on top of the others, (3) a gold nanoparticle sits approximately at the center of a AAO hexagonal cell, (4) no more than two gold nanoparticles occupy a single AAO hexagonal cell.

In most cases, the gold nanoparticles formed a monolayer in small domains while in other small domains they were absent. This is probably due to the lack of uniform opening of the alumina barrier layer to expose the filled Au (the process of Figure 1e). In other words, the gold nanoparticles form a monolayer on well-opened domains of AAO, while closed AAO domains contain no nanoparticles. Once we were very careful about the uniformity of etching, the gold nanoparticle coverage can be as high as about 0.9 over an area of almost 1 cm<sup>2</sup>. We also tried this experiment with Ag instead of Au, and alkyl thiols instead of 1,4-phenylene diisocyanide. The general trend of above holds for these alternative systems.

In summary, we have developed the methodology to produce nanoscale gold rods using an AAO template. Each gold rod was generated in every AAO pore. This nanoislands array of gold formed over the AAO pores can be used as corner stones for building nanostructures. We demonstrated this by forming a nanostructure on the Au/AAO by binding a self-assembly class of molecules onto the metal islands.

## **Experimental Section**

To synthesize the AAO template, an aluminum sheet (99.999%, 0.1 mm thick) was anodized under 40 V DC relative to a counter electrode (graphite) in 0.3 M oxalic acid maintained at 0 °C. The first anodization process was for 10 hours. After etching with a solution of 1.8% chromic acid and 6% phosphoric acid maintained at 60 °C for 10 hours, the second anodization process was carried out for 1 hour.

For the barrier layer thinning, the DC voltage in the second anodization process is gradually lowered to less than 10 V until no current is read. As the current rises again to about 5 mA in a few minutes, the DC voltage is dropped again for no current. As this process is repeated, the time interval for the voltage drop increases and the degree of the drop decreases. As the voltage drop is smooth in an exponential fashion over the time, rather uniform thinning can be achieved without undesired dendritic structure. We performed this thinning process by the gradual voltage drop for about 20 minutes.

Acknowledgements. This work was supported by National R&D Project for Nano Science and Technology (Contract No. M10214000110-02B1500-01800) of MOST and SRC program (Center for Nanotubes and Nanostructured Composites) of KOSEF.

## References

- 1. Shingubara, S. J. Nanoparticle Res. 2003, 5, 17.
- (a) Masuda, H.; Fukuda, K. Science **1995**, 268, 1466. (b) Masuda, H.; Satoh, M. Jpn. J. Appl. Phys. **1996**, 35, L126. (c) Masuda, H.; Yotsuya, M.; Ishida, M. Jpn. J. Appl. Phys. **1998**, 37, L1090. (d) Masuda, H.; Yasui, K.; Nishio, K. Adv. Mater. **2000**, 12, 1031.
- (a) Routkevitch, D.; Tager, A. A.; Haruyama, J.; Almawlawi, D.; Moscovits, M.; Xu, J. M. *IEEE Transactions on Electron Devices* **1996**, *43*, 1646. (b) Li, J.; Padadopoulos, C.; Xu, J. M.; Moscovits, M. *Appl. Phys. Lett.* **1999**, *75*, 367.
- (a) Hulteen, J. C.; Martin, C. R. J. Mater. Chem. 1997, 7, 1075. (b) Martin, C. R. Chem. Mater. 1996, 8, 1739.
- (a) Suh, J. S.; Lee, J. S.; Kim, H. Syn. Met. 2001, 123, 381. (b) Lee, J. S.; Gu, G. H.; Kim, H.; Suh, J. S.; Han, I.; Lee, N. S.; Kim, J. M.; Park, G.-S. Syn. Met. 2001, 124, 307.
- (a) Kim, M. J.; Lee, T. Y.; Choi, J. H.; Park, J. B.; Lee, J. S.; Kim, S. K.; Yoo, J.-B.; Park, C.-Y. *Diamond and Related Mater.* 2003, *12*, 870. (b) Kim, M. J.; Choi, J. H.; Park, J. B.; Kim, S. K.; Yoo, J.-B. *Thin Solid Films* 2003, *435*, 312.
- (a) Masuda, H.; Yamada, H.; Satoh, M.; Asoh, H.; Nakao, M.; Tamamura, T. *Appl. Phys. Lett.* **1997**, *71*, 2770. (b) Masuda, H.; Asoh, H.; Watanabe, M.; Nishio, K.; Nakao, M.; Tamamura, T. *Adv. Mater.* **2001**, *13*, 189. (c) Pang, S. W.; Tamamura, T.; Nakao, M.; Ozawa, A.; Masuda, H. *J. Vac. Sci. Technol. B* **1998**, *16*, 1145.
- Shingubara, S.; Okino, O.; Sayama, Y.; Sakaue, H.; Takahagi, T. Jpn. J. Appl. Phys. 1997, 36, 7791.
- (a) Choi, J.; Schilling, J.; Nielsch, K.; Hillebrand, R.; Reiche, M.; Wehrspohn, R. B.; Gösele, U. *Mater. Res. Soc. Symp. Proc.* 2002, 722, L5.2. (b) Choi, J.; Nielsch, K.; Reiche, M.; Wehrspohn, R. B.; Gösele, U. *J. Vac. Sci. Technol. B* 2003, 21, 763.
- Furneaux, R. C.; Rigby, W. R.; Davidson, A. P. Nature 1989, 337, 147.
- Nielsch, K.; Müller, F.; Li, A. P.; Gösele, U. Adv. Mater. 2000, 12, 582.
- Choi, J.; Sauer, G.; Nielsch, K.; Wehrsphon, R. B.; Gösele, U. Chem. Mater. 2003, 15, 776.
- (a) Kim, H. S.; Lee, S. J.; Kim, N. H.; Yoon, J. K.; Park, H. K.; Kim, K. Langmuir 2003, 19, 6701 (b) Robertson, M. J.; Angelici, R. J. Langmuir 1994, 10, 1488 (c) Irwin, M. J.; Jia, G.; Payne, N. C.; Puddephatt, R. J. Organometallics 1996, 15, 51 (d) Henderson, J. I.; Feng, S.; Ferrence, G. M.; Bein, T.; Kubiak, C. P. Inorg. Chem. Acta 1996, 242, 115 (e) Lin, S.; McCarley, R. L. Langmuir 1999, 15, 151 (f) Henderson, J. I.; Feng, S.; Bein, T.; Kuboak, C. P. Langmuir 2000, 16, 6183 (g) Horswell, S. L.; O'Neil, I. A.; Schiffirin, D. J. J. Phys. Chem. B 2001, 105, 941.